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1 **Electrospun pH-sensitive core-shell polymer nanocomposites**
2 **fabricated using a tri-axial process**

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41 **Abstract:**

42 A modified tri-axial electrospinning process was developed for the generation of a
43 new type of pH-sensitive polymer/lipid nanocomposite. The systems produced are
44 able to promote both dissolution and permeation of a model poorly water-soluble drug.
45 First, we show that it is possible to run a tri-axial process with only one of the three
46 fluids being electrospinnable. Using an electrospinnable middle fluid of Eudragit
47 S100 (ES100) with pure ethanol as the outer solvent and an unspinnable
48 lecithin-diclofenac sodium (PL-DS) core solution, nanofibers with linear morphology
49 and clear core/shell structures can be fabricated continuously and smoothly. X-ray
50 diffraction proved that these nanofibers are structural nanocomposites with the drug
51 present in an amorphous state. *In vitro* dissolution tests demonstrated that the
52 formulations could preclude release in acidic conditions, and that the drug was
53 released from the fibers in two successive steps at neutral pH. The first step is the
54 dissolution of the shell ES100 and the conversion of the core PL-DS into sub-micron
55 sized particles. This frees some DS into solution, and later the remaining DS is
56 gradually released from the PL-DS particles through diffusion. *Ex vivo* permeation
57 results showed that the composite nanofibers give a more than two-fold uplift in the
58 amount of DS passing through the colonic membrane as compared to pure DS; 74%
59 of the transmitted drug was in the form of PL-DS particles. The new tri-axial
60 electrospinning process developed in this work provides a platform to fabricate
61 structural nanomaterials, and the core-shell polymer-PL nanocomposites we have
62 produced have significant potential applications for oral colon-targeted drug delivery.

63 **Keywords:** Tri-axial electrospinning; core-sheath fibers; polymer-lipid
64 nanocomposites; colon-targeted drug delivery; electrospinnability

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70 **1. Introduction**

71 The fabrication of advanced drug delivery systems (DDSs) is increasingly
72 dependent on the creation of complex architectures and understanding
73 structure-activity relationships at the nanoscale [1-3]. To this end, core-shell
74 nanostructures have been very widely studied in the production of functional
75 nanomaterials, including those for biomedical applications [4-6]. For drug delivery
76 and controlled release, both the core and shell can be loaded with an active
77 pharmaceutical ingredient (API) and/or with different types of pharmaceutical
78 excipients. Applications of such systems include improving the solubility of poorly
79 water-soluble drugs, controlled release of multiple APIs from a single dosage form, or
80 tunable multiple phase release [7-9].

81 Over recent years, polymers and lipids have been the most widely used
82 pharmaceutical excipients, and these materials have acted as the basis for a broad
83 gamut of novel DDSs, being exploited to alter the biopharmaceutical and
84 pharmacokinetic properties of the drug molecule for favorable clinical outcomes
85 [3,10,11]. Numerous core-shell polymeric nanoparticles (NPs) and lipid-based DDS
86 (such as solid lipid dispersions and liposomes) have been investigated for drug
87 delivery through varied administration routes [12-15]. Novel strategies derived from
88 the combined usage of polymers and phospholipids (PLs) have been reported for
89 some biomedical applications (including controlled release) and are presently of
90 intense interest in the pharmaceuticals field. However, virtually all the reported
91 polymer-lipid composites are in the form of microparticles or NPs [4,8,16-18].

92 Core-shell nanofiber-based DDS have received relatively little attention, and to the
93 best of our knowledge there are no reports of drug-loaded polymer-lipid nanofibers
94 being used in drug delivery.

95 Electrospun nanofibers, comprising an API loaded into a filament-forming
96 polymer, have been the focus of much research. They are prepared from a
97 co-dissolving solution of a drug and polymer; this is ejected from a syringe with
98 electrical energy used to rapidly evaporate the solvent and yield one-dimensional
99 fibers with diameters frequently on the nanoscale. This technique is scalable, and
100 several recent reports address large scale fabrication and the potential for commercial
101 products [19-22]. The intense research effort invested in these materials thus appears
102 to be about to yield products which can make a major difference to patients' lives.
103 Electrospinning is a facile, one-step procedure, and the products form as a visible and
104 flexible mat which can easily be recovered from the collector without significant loss
105 of material or damage. The nanofibers produced can further be used as templates to
106 manipulate molecular self-assembly to create drug-loaded NPs or liposomes; the
107 electrospinning technique thus provides not only a bridge between fiber-based and
108 NP-based DDSs, but also between solid and liquid dosage forms [23-26].

109 The most simple, single-fluid, electrospinning process has been explored for
110 approaching two decades, and the applications of the resultant monolithic nanofibers
111 have been probed in a wide range of fields. Current developments in electrospinning
112 are focused in two key areas. The first is the manufacture of electrospun nanofibers on
113 an industrial scale [27-29]. The second line of research involves developing advanced

114 electrospinning techniques to yield nanofibers with sophisticated structural
115 characteristics (such as multiple-compartment nanofibers, core-shell nanofibers, or
116 structured fibers with varied distributions of the API), which in turn impart tunable
117 and multiple functionalities [30-32]. Because of the popularity of core-shell
118 nanostructures and the relative ease of the process, coaxial electrospinning (in which
119 two needles, one nested inside another, are used to handle two working fluids) has
120 been the focus of much research. Other advanced approaches such as side-by-side
121 electrospinning (to yield Janus fibers), tri-axial electrospinning (giving three-layer
122 composites), and other types of multiple-fluid electrospinning have been neglected in
123 comparison [6,9,33].

124 Compared with single-fluid electrospinning, the standard coaxial experiment has
125 greatly expanded the range of fibers which can be produced. These include not only
126 core-shell fibers [34,35], but also fibers prepared from materials without
127 filament-forming properties [36] and used as templates for creating nanotubes (from
128 the fiber as a whole) or the “bottom-up” generation of NPs (self-assembled from the
129 components loaded in the fibers) [26, 37]. For biomedical applications, core-shell
130 nanofibers proffer a series of new possibilities; for instance, it is possible to protect a
131 fragile active ingredient such as a protein from the stresses of the electrospinning
132 processes by confining it to the core, or to vary the APIs concentration in the core and
133 shell to achieve complex drug release profiles [38-41]. In the traditional coaxial
134 process the sheath working fluid must be electrospinnable, but a modified process in
135 which one can utilize unspinnable liquids as the sheath fluid is also possible. The

136 number of polymers which can be directly electrospun is rather limited, but there are
137 numerous unspinnable liquids, and the modified coaxial process should hence further
138 expand the range of functional nanofibers which can be produced [38,42,43].

139 The above discussion is focused on the simultaneous processing of two fluids;
140 working with three or even four fluids simultaneously is also possible, however
141 [44-49]. For example, Han and Steckl reported tri-layer nanofibers for biphasic
142 controlled release, using dyes as model active ingredients [49]. In very recent work,
143 we successfully developed a tri-axial electrospinning process to generate nanofibers
144 with a gradient distribution of the API, allowing us to achieve zero-order drug release
145 profiles [31]. However, in all the tri-axial electrospinning processes reported to date,
146 the three working fluids are all electrospinnable. This limits the applications of the
147 process. If unspinnable liquids can be processed in combination with spinnable
148 working solutions, a much broader selection of functional products could be designed
149 and generated.

150 Building on our previous work developing modified coaxial [38,42,43] and
151 standard tri-axial electrospinning [50], here we report the first modified tri-axial
152 electrospinning process. We have used this process to create core-shell fibers
153 comprising a lipid-drug core and a pH sensitive shell, thereby allowing us to
154 demonstrate that only an electrospinnable central fluid is required to achieve a
155 successful tri-axial process. The polymer-lipid nanocomposites produced showed
156 desirable functional performance in altering the release behavior of the model drug
157 diclofenac sodium and improving its permeation through the colonic membrane.

158 **2 Experimental**

159 **2.1. Materials**

160 Eudragit S100 (ES100, $M_w=135,000$), a methacrylic acid/methyl methacrylate
161 copolymer which only dissolves at $\text{pH} > 7.0$, was obtained from Röhm GmbH
162 (Darmstadt, Germany). Diclofenac sodium (DS, a non-steroidal anti-inflammatory
163 drug with potent anti-inflammatory, analgesic and antipyretic properties) was
164 purchased from the Hubei Biocause Pharmaceutical Co., Ltd. (Hubei, China).
165 Lecithin (PL, extracted from egg yolk, and containing lysophosphatidylcholine,
166 sphingomyelin, and neutral lipids in minor quantities), N,N-dimethylacetamide
167 (DMAc), anhydrous ethanol, methylene blue and basic fuchsin were purchased from
168 the Sinopharm Chemical Reagent Co., Ltd. (Shanghai, China). All other chemicals
169 used were analytical grade, and water was doubly distilled before use.

170 **2.2. Electrospinning**

171 The tri-layer concentric spinneret was homemade. Three syringe pumps
172 (KDS100, Cole-Parmer, Vernon Hills, IL, USA) and a high-voltage power supply
173 (ZGF 60kV/2 mA, Shanghai Sute Corp., Shanghai, China) were used for
174 electrospinning. The collector comprised a flat piece of cardboard wrapped with
175 aluminum foil. All electrospinning processes were carried out under ambient
176 conditions (21 ± 5 °C with a relative humidity of 47 ± 5 %). Experiments were
177 recorded using a digital camera (PowerShot A490, Canon, Tokyo, Japan). The
178 spinneret to collector distance was fixed at 15 cm for all experiments.

179 The outer fluid was pure anhydrous ethanol. The middle fluid consisted of 14.0 g

180 ES100 in 100 mL of a mixture of ethanol / DMAc (90:10 v/v). The inner fluid was
181 prepared from 3 g PL and 0.6 g DS in 10 mL ethanol. After initial optimization
182 experiments, the applied voltage was fixed at 15 kV. To facilitate observation of the
183 electrospinning processes, 10 mg/L methylene blue was added to the inner fluid and 5
184 mg/L basic fuchsin to the middle fluid. Four different sets of fibers were prepared
185 with varied flow rates, as detailed in Table 1.

186 **Table 1.**

187 **2.3. Characterization**

188 *2.3.1. Morphology*

189 The morphology of the fibers was determined using a Quanta FEG450 field
190 emission scanning electron microscope (FESEM; FEI Corporation, Hillsboro, OR,
191 USA). Prior to examination, samples were gold sputter-coated under a nitrogen
192 atmosphere to render them electrically conductive. Images were recorded at an
193 excitation voltage of 20 kV. The average fiber size was determined by measuring their
194 diameters at more than 100 places in FESEM images, using the NIH Image J software
195 (National Institutes of Health, MD, USA). To view the cross-sections of sample F2, a
196 section of the fiber mat was placed into liquid nitrogen and manually broken before
197 gold coating.

198 Transmission electron microscope (TEM) images of the samples were recorded
199 on a JEM 2100F field emission TEM (JEOL, Tokyo, Japan). Samples were collected
200 by fixing a lacey carbon-coated copper grid on the collector and electrospinning
201 directly onto it for several minutes.

202 2.3.2. *Physical form and compatibility*

203 X-ray diffraction (XRD) was conducted using a D/Max-BR diffractometer
204 (Rigaku, Tokyo, Japan) with Cu K α radiation over the 2θ range 5 to 60° at 40 kV and
205 30 mA. Attenuated total reflectance-Fourier transform infrared (ATR-FTIR)
206 spectroscopy was carried out on a Nicolet-Nexus 670 FTIR spectrometer (Nicolet
207 Instrument Corporation, Madison, USA) from 500 cm⁻¹ to 4000 cm⁻¹ at a resolution
208 of 2 cm⁻¹.

209 2.3.3. *In vitro dissolution tests*

210 To determine drug loading efficiency (LE), 0.100 g of the fibers was added into
211 10 mL of a 10% v/v ethanol solution in water, in order to extract all the loaded DS.
212 The resultant solutions were diluted using phosphate buffered saline (PBS, pH7.0,
213 0.1M) to a suitable concentration for UV measurement. The LE was calculated using
214 the following equation:

$$215 \quad \text{LE(\%)} = (\text{DS mass measured})/(\text{theoretical DS mass in the formulation}) \times 100\%$$

216 *In vitro* dissolution tests were carried out according to the Chinese
217 Pharmacopoeia (2015 Ed.). Method II, which is a paddle method, was undertaken
218 using a RCZ-8A dissolution apparatus (Tianjin University Radio Factory, Tianjin,
219 China). 280 mg of fibers F2 or 20 mg of the DS raw material (particle size <30 μm)
220 were first placed in 600 mL of 0.1 M HCl. Two hours later, 2.4 g NaOH was added to
221 neutralize the dissolution media. The temperature of the dissolution medium was $37 \pm$
222 1 °C and the instrument was stirred at 50 rpm. Sink conditions were maintained, with
223 $C < 0.2C_s$. At predetermined time points, 5.0 mL aliquots were withdrawn from the

224 dissolution medium and replaced with distilled water to maintain a constant volume.
225 After filtration through a 0.22 μm membrane (Millipore, Billerica, MA, USA) and
226 appropriate dilution with PBS, samples were analyzed at $\lambda_{\text{max}} = 276 \text{ nm}$ using a
227 UV-vis spectrophotometer (UV-2102PL, Unico Instrument Co. Ltd., Shanghai,
228 China). The cumulative amount of DS released at each time point was back-calculated
229 from the data obtained against a predetermined calibration curve. Experiments were
230 performed seven times, and the average results from six of these replicates are
231 reported as mean \pm S.D.

232 During the *in vitro* dissolution process, dissolution media from the seventh
233 replicate was withdrawn and the transmittance at $\lambda=500 \text{ nm}$ measured using the
234 UV-vis spectrophotometer. The average hydrodynamic diameter and size distribution
235 of the particles in the final dissolution medium from these experiments were
236 determined using a BI-200SM static and dynamic light scattering (SDLC) instrument
237 (Brookhaven Instruments Corporation, Austin, TX, USA).

238 2.3.4. *Ex vivo* permeation tests

239 *Ex vivo* permeation studies were performed using a RYJ-6A diffusion test
240 apparatus (Shanghai Huanghai Drug Control Instrument Co., Ltd., Shanghai, China),
241 in which materials were mounted in six Keshary-Chien glass diffusion cells and a
242 water bath system maintained a constant temperature of $37 \pm 0.2 \text{ }^\circ\text{C}$. Each cell had a
243 diffusion area of 2.60 cm^2 , and the receptor compartment had a capacity of 7.2 mL
244 PBS (pH7.0, 0.1M). Each donor compartment was filled with 1.0 mL PBS and the
245 hydrodynamics in the receptor compartment were maintained by stirring at 50 rpm

246 with a Teflon coated magnetic bead. Large intestines were obtained from pigs after
247 slaughtering (Baoshan Jiangwan slaughterhouse, Shanghai, China). The intestine was
248 washed carefully with physiological saline solution (NaCl 0.9% w/v) to remove
249 non-digested food. The colonic membranes were peeled away from the intestines and
250 fixed on diffusion cells with the mucosal walls upward. They were equilibrated at
251 35 °C for 30 min before permeation tests.

252 The F2 fibers (140 mg) were placed on the mucosal surface in the chambers.
253 Samples (1 mL) were withdrawn from the receptor compartment at timed intervals
254 and 1 mL fresh PBS was added to maintain the volume of fluid here at a constant
255 level. The aliquots were filtered through a 0.22 μm membrane (Millipore, Billerica,
256 MA, USA). The absorption of the filtrate was measured at 276 nm to determine the
257 amount of DS present in the aqueous phase. The semi-solid residue was dissolved
258 using 10 mL of a 10% v/v ethanol solution in water and diluted with PBS before
259 measuring absorbance, in order to determine its DS content of. All measurements
260 were carried out in triplicate. Permeation experiments with 10 mg of pure DS (particle
261 size $<30 \mu\text{m}$) as a control were also carried out.

262 2.4. Statistical analysis

263 The experimental data are presented as mean \pm SD. The results from the *in vitro*
264 dissolution tests and *ex vivo* permeation tests were analyzed using one-way ANOVA.
265 The threshold significance level was set at 0.05. Thus, *p* (probability) values lower
266 than 0.05 were considered statistically significant.

267 3. Results and discussion

268 3.1. Implementation of modified tri-axial electrospinning

269 A diagram illustrating the modified tri-axial electrospinning process is shown in
270 Fig. 1. The system consists of four components: three syringe pumps to drive the
271 working fluids, a power supply, a fiber collector, and a three layer concentric
272 spinneret. In modified coaxial electrospinning, the use of a spinnable core solution
273 can ensure a successful process regardless of the electrospinnability of the sheath fluid
274 [43]. Here, the central solution is electrospinnable, and this is utilized to achieve
275 tri-axial electrospinning even though the outer fluid is pure solvent and the inner fluid
276 is unspinnable.

277 **Fig. 1**

278 The homemade tri-concentric spinneret and its connection with the power supply
279 and three working fluids are shown in Fig. 2a. An alligator clip was used to connect
280 the power supply to the spinneret, which was directly fixed to the syringe holding the
281 outer fluid. The middle and inner fluids were connected to the spinneret through
282 high-elastic silicon tubing.

283 The design of the spinneret is of critical importance in ensuring a robust and
284 reproducible electrospinning process [42,48]. A well-designed spinneret must provide
285 a suitable template for producing the desired nanofiber architectures, and must be
286 developed bearing in mind the behavior of the working fluids under an electrical field.
287 The spinneret used in this work is exhibited in the top-right and bottom-right insets of
288 Fig. 2a. It consists of three concentric capillaries composed of austenitic stainless steel
289 ($O_6Cr_{19}Ni_{10}$, GB24511 in China). The inner, middle and outer capillaries have outer
290 diameters of 0.4, 1.6, 2.8 mm and inner diameters (D_i) of 0.20, 1.3 and 2.2 mm,

291 respectively. The end of the inner capillary projects 0.2 mm out of the central one,
292 which similarly projects 0.2 mm from the outer capillary. This design helps to ensure
293 the encapsulation of the inner fluid by the middle fluid, and in turn the middle by the
294 outer fluid. This structure should also help to prevent mixing of the working fluids
295 when they are ejected from the spinneret.

296 **Fig. 2**

297 Under optimised conditions (see Section 2.2), successful electrospinning could
298 be achieved as shown in Fig. 2b. The process involves three steps including Taylor
299 cone formation, the emission of a straight fluid jet and then an unstable region with
300 gradually enlarged bending and whipping loops. The top-right inset of Fig. 2b
301 displays the droplets ejected from the spinneret with no voltage applied. Both the blue
302 inner fluid and pink middle fluids were observed to diffuse into the outer fluid to
303 some extent, as demonstrated by their gradually increased sizes and decreased size of
304 the outer (colourless) solvent moving away from the spinneret. However, the three
305 working fluids form a clear three-layer compound Taylor cone when a voltage of 15
306 kV was applied, as shown in the bottom-right inset of Fig. 2b.

307 The modified tri-axial electrospinning process can be run continuously and
308 smoothly, without any clogging or other adverse phenomena arising. These are
309 frequently encountered in traditional single-fluid and coaxial electrospinning [50], but
310 spinning with a pure solvent as the exterior fluid has been shown to reduce incidents
311 of clogging as well as to improve the uniformity of the fibers produced in the latter
312 process [42]. The use of pure solvent as the outer layer will: 1) lubricate the spinneret

313 to retard clinging; 2) prevent the formation of semi-solid substance on the surface of
314 the fluid jets; 3) protect the inner fluid from any environmental fluctuations; and, 4)
315 lead to a longer drawing period under the electrical field, and thus to narrower fibers.

316 *3.2. Morphology and core-shell structures of the created nanofibers*

317 FESEM images of fibers F1 to F4 are shown in Fig. 3. When the inner and
318 outer fluids were turned off, a traditional single-fluid electrospinning of the middle
319 ES100 solution could be achieved. Although manual intervention was needed
320 periodically to remove semi-solid substances which collected on the spinneret, the
321 resultant ES100 fibers were linear without any beads-on-a-string or
322 spindles-on-a-string morphology (Fig. 3a1 and 3a2). These fibers have an average
323 diameter of $1.27 \pm 0.13 \mu\text{m}$, with an uneven and wrinkled surface (Table 1, Fig. 3a2).
324 This is a result of barometric pressure, when residual solvent which was not
325 evaporated during electrospinning escaped from the fibers. Single-fluid
326 electrospinning easily traps solvent in the fibers because of the formation of a solid
327 “skin” on the fluid jet during the solidification process.

328 The F2 fibers are linear with an average diameter of $0.55 \pm 0.06 \mu\text{m}$ and smooth
329 surfaces (Fig. 3b1 and 3b2, Table 1). This can be attributed to the surrounding outer
330 solvent and appropriate selection of the flow rates of the three working fluids (0.5, 2.0
331 and 0.5 mL/h for outer, middle and inner fluids, respectively). If the flow rate of the
332 outer solvent is kept constant and those of the middle and inner fluid altered to 1.6
333 and 0.9 mL/h respectively, the resultant F3 material has many beads clinging to the
334 fibers, although the latter are still linear with an average size of $0.47 \pm 0.05 \mu\text{m}$ (Fig.

335 3b1 and 3b2, Table 1). It is thought that this high flow rate of the inner fluid causes it
336 to penetrate the middle and outer fluids to form round PL-DS beads on the fiber
337 surfaces.

338 If the flow rate of the outer solvent is doubled to 1.0 mL/h, the fibers generated
339 exhibit a typical spindles-on-a-string morphology (Fig. 3d1 and 3d2). Some
340 unexpected clumps are also formed within the fiber mat, as shown in the inset of Fig.
341 3d1. These are ascribed to PL escaping from the inner fluids. A further increase of the
342 outer solvent flow rate was found to result in an electrospaying process. These results
343 demonstrate that the selection of flow rates is a key parameter which must be
344 controlled to ensure the formation of a core-shell nanostructure.

345 **Fig. 3**

346 FESEM images of cross-sections of F2 (Fig. 4a) and TEM images (Fig. 4b)
347 demonstrate that the fibers have clear core/shell structures. Both the FESEM and
348 TEM images suggest that the PL-DS core has a diameter of approximately 300 nm.

349 **Fig. 4**

350 *3.3. Physical form and component compatibility*

351 XRD data are depicted in Fig. 5; these clearly demonstrate that DS is crystalline,
352 with many sharp Bragg reflections visible in its pattern. ES100 exhibits only a broad
353 hump, characteristic of an amorphous material. PL exists as a paste at an ambient
354 temperature of 21 °C, yet shows a sharp reflection at $2\theta=5.18^\circ$. This suggests that
355 there are liquid crystals present in the PL paste, with an ordered lamellar structure as
356 reported in the early literature [51]. All reflections from PL and DS are absent in the

357 patterns of the core-shell F2 fibers, suggesting the formation of an amorphous PL-DS
358 complex.

359 **Fig. 5**

360 The potential secondary interactions between the fiber components were
361 investigated using ATR-FTIR, and the results are shown in Fig. 6. DS has three
362 characteristic peaks at 1574, 1553 and 1507 cm^{-1} arising from its benzene rings. In the
363 spectrum of PL, the CH_2 symmetric and asymmetric vibrations at 2854 cm^{-1} and 2923
364 cm^{-1} and the antisymmetric stretch of $\text{N}^+(\text{CH}_3)_3$ at 968 cm^{-1} comprise the most
365 prominent features. These peaks similarly appear in the spectrum of the fibers,
366 confirming the presence of PL with ES100. However, the characteristic peaks from
367 the benzene rings of DS cannot be seen in the F2 spectrum. This can be attributed to
368 secondary interactions between PL and DS. Hydrophobic interactions, in addition to
369 possible hydrogen bonding and electrostatic interactions, can arise between all three
370 components in F2, as is clear from a consideration of the molecular structures in Fig.
371 6. These secondary interactions should ensure that the drug and excipients are highly
372 compatible, favorable for the stability of the core-shell nanocomposites.

373 **Fig. 6**

374 *3.4. Functional performance*

375 DS has a maximum absorbance at 276 nm, which was used to construct a
376 calibration curve: $A=0.0085+0.0279 C$ ($R=0.9997$) within a linear range from 2 to 50
377 $\mu\text{g/mL}$. The drug content in F2 was first assayed, and found to be $7.26 \pm 0.31\%$ ($n =$
378 6), almost identical to the calculated value of 7.14%.

379 The *in vitro* release profiles of F2 and the DS starting material are shown in Fig.
380 7a. DS is virtually insoluble in acidic conditions, with a small increase in solubility
381 when the pH is raised to neutral. After 2 h in acid, 2.8% of DS from the raw material
382 was freed into the dissolution media. When the pH was raised to neutral, the DS
383 particles gradually dissolved over *ca.* 3 hours. For F2, 2.1% of the loaded DS was
384 released during the first 2 h. In the neutral dissolution media, the nanofibers released a
385 total of 79.1% of the incorporated DS over 22 h.

386 ES100 is a pH-sensitive polymer, and is insoluble at pHs below 7.0; it can thus
387 be used to target DS to the colon region. DS is a popular API for oral administration
388 and is frequently used in the treatment of pain and peri-operatively. However, it can
389 easily result in an anaphylactic reaction, and to an allergic reaction in the digestive
390 tract [52,53]. With traditional electrospun nanofibers, the drug is released by diffusion
391 through an insoluble polymeric matrix, or by an erosion mechanism from a
392 water-soluble carrier (or a combination of both processes) [39,43]. Here the drug
393 release from the core-shell composites is expected to include two successive steps
394 (Fig. 7b). First, dissolution of the pH-sensitive ES100 shell will occur, with some
395 diffusion of DS from the insoluble core PL. After dissolution of the shell ES100, the
396 core PL-DS is not thought to be able to endure the shear forces of stirring applied
397 during the experiment and thus we propose that the core is broken up into PL-DS
398 particles. The DS is then gradually released from the resultant DS-PL aggregates.
399 Thus in the dissolution tests, the released drug (%) corresponds to the DS molecules
400 which are in solution (the DS-PL aggregates in suspension are removed by filtration).

401

Fig. 7

402 To further investigate the drug release mechanism and validate this hypothesis,
403 the transmittance of the dissolution media and light scattering studies were performed.
404 The changes in transmittance at $\lambda=500$ nm are shown in Fig. 8a. DS has no
405 absorbance above 320 nm, and thus any turbidity of the dissolution media recorded at
406 this wavelength must result from the formation of a PL-DS suspension. In the first 2 h,
407 the transmittance remains virtually constant. After the pH is raised to neutral, the
408 transmittance values decreased for 3 h, after which they level out at around 77%. This
409 is consistent with the dissolution of the shell ES100 occurring over this period and
410 resulting in PL-DS nanoparticles.

411 The SDLC results obtained on the final dissolution medium are given in Fig. 8b.
412 The PL-DS particles formed have an average diameter of 434 nm with a
413 polydispersity index (PDI) of 0.187

414

Fig. 8

415 The results of permeation tests are presented in Fig. 9. DS is a Biopharmaceutical
416 class II drug, meaning it is poorly water-soluble but is able to effectively cross fatty
417 membranes [54]. After 12 h only 3.7 mg of the pure drug was transmitted into the
418 receptor cells. The dissolution of DS is very slow because there is only a very limited
419 amount of aqueous medium in the donor cell (*cf.* the dissolution experiments, which
420 are performed under sink conditions) [55]. For the F2 fibers, both dissolved DS and
421 the PL-DS particles penetrate the bio-membranes into the receptor cells [56].
422 Although the core-shell nanofibers provided sustained release of DS in dissolution

423 studies (much slower than the release from pure DS), after 12 h 8.1 ± 0.46 mg DS
424 from F2 had entered the receptor chamber. Of this amount, 1.7 ± 0.23 mg was present
425 in the aqueous phase (or in particles below 220 nm in size, which could pass through
426 the filters used). This suggests that $(8.1-1.7)/8.1 \times 100 = 79\%$ of the DS penetrated
427 through the mucosal membrane in the format of PL-DS particles. For oral
428 administration applications, this drug delivery route should alleviate any potential
429 allergic reactions with the digestive tracts.

430 Many commercial tablets are essentially a physical mixture of drug powders and
431 polymeric carrier, the latter being added to modulate the drug release behavior. The
432 combined use of polymer and lipid in the fibers prepared in the work is able to both
433 protect the API from release in the stomach and provide sustained release in the
434 colonic region, and also ensure improved trans-membrane permeability, leading to
435 more effective absorbance. This strategy is particularly useful for oral delivery of
436 Class IV drugs (which are both poorly water-soluble and have poor permeation
437 properties). Drug-loaded electrospun fibers can easily be converted into routine oral
438 solid dosage forms such as tablets and capsules using traditional pharmaceutical
439 protocols [57-59].

440 **Fig. 9**

441 *3.5. Perspectives*

442 Coaxial electrospinning is often regarded as a major breakthrough in this field
443 [60,61]. The fact that only one of the working fluids needs to be electrospinnable for a
444 successful coaxial process significantly widens the range of materials which can be

445 processed, and a very broad family of core-shell nanostructures can be produced.
446 There are only about 100 polymers which can be directly electrospun into fibers, and
447 often these can only be processed within a narrow window of conditions
448 (concentration, voltage, etc). The introduction of unspinnable fluids in the modified
449 coaxial processes greatly expands the capability of this simple technology to produce
450 nanoscale products from a large range of raw materials. Furthermore, modified
451 coaxial electrospinning permits all types of liquid phase (including solvents, small
452 molecule solutions, dilute polymer solutions, suspensions and also emulsions) to be
453 processed.

454 In this work, we report the first example of modified tri-axial electrospinning.
455 Similar to modified co-axial spinning, this moves technology beyond the traditional
456 tri-axial process in which all three working fluids are required to be individually
457 electrospinnable. In our work, two of the three fluids were unspinnable alone: an
458 electrospinnable middle layer fluid is sufficient to ensure a successful tri-axial process.
459 This proof-of-concept work indicates that there are many possibilities in developing
460 functional nanofibers through the introduction of unspinnable outer-layer and
461 inner-layer working fluids into tri-axial processes.

462 The feasibility of the different tri-axial electrospinning processes which can be
463 conceived are summarized in Fig. 10. A process with three spinnable working fluids
464 (Process I) has been reported in several publications [31,44,45]. Processes II, III and
465 IV have two of the three fluids being electrospinnable, and these are feasible provided
466 the working fluids are compatible. This report is an example of Process V, with a

467 spinnable middle layer fluid used to support unspinnable outer and inner fluids. For
468 processes VI and VII, the two unspinnable fluids are adjacent to each other. This may
469 result in diffusion of the solutes and formation of a mixture of the two unspinnable
470 liquids, and thus it is anticipated that such to tri-axial electrospinning processes will
471 result in failure.

472 **Fig. 10**

473

474 **4. Conclusions**

475 A modified tri-axial electrospinning process was successfully implemented to create
476 core-shell nanofibers, in which a spinnable Eudragit S100 (ES100) solution was used
477 as the middle fluid to support the outer solvent and an unspinnable phosphatidyl
478 choline (PL)/diclofenac sodium (DS) inner solution. This resulted in a continuous and
479 trouble-free nanofabrication process. The resultant core-shell nanofibers have a linear
480 morphology with an obvious core-shell structure. XRD demonstrated that the
481 nanofibers are structural nanocomposites with both the drug DS and also the lipid
482 carrier PL losing their original crystalline physical forms and being transferred into an
483 amorphous state. These core (PL-DS)-shell (ES100) nanostructures can protect the
484 drug from release in acidic conditions to give colon-targeted release. They release the
485 drug through two successive steps at neutral pH: first, dissolution of the shell ES100
486 occurs, which is believed to generate PL-DS sub-micron sized particles. Subsequently,
487 release of DS from the particles occurs. The composite nanofibers lead to more than
488 twice as much drug permeation through the colonic bio-membrane when compared

489 with pure DS. The tri-axial electrospinning process developed in this work should
490 provide a new platform to fabricate structural nanomaterials, and polymer-PL
491 nanocomposites such as those prepared here can be utilized for effective oral drug
492 delivery.

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672

673

674

675 **Table and Figures Legend**

676

677 **Table 1.** Key details of the electrospinning processes and resultant fibers

678 **Fig. 1.** A diagram of the modified tri-axial electrospinning process and its use for
679 preparing core-shell drug-loaded nanofibers.

680 **Fig. 2.** The implementation of modified tri-axial electrospinning: (a) the connection of
681 the spinneret with the power supply and the working fluids (left), and images of the
682 spinneret (insets); (b) a digital photograph of the tri-axial process (left), the droplet
683 before a voltage of 15 kV was applied (top-right) and the compound Taylor cone
684 (bottom-right).

685 **Fig. 3.** FESEM images of the core-shell nanofibers and their size distributions; (a1
686 and a2) F1; (b1 and b2) F2; (c1 and c2) F3; (d1 and d2) F4. The inset to (d1) shows a
687 clump of PL-DS.

688 **Fig. 4.** (a) A FESEM image of the cross-sections of F2 and (b) a TEM image showing
689 the same.

690 **Fig. 5.** XRD patterns of the raw materials (PL, EL100 and DS) and F2.

691 **Fig. 6.** ATR-FTIR spectra and the molecular formula of the fiber components.

692 **Fig. 7.** *In vitro* dissolution of DS and D2 (a) and the proposed drug release
693 mechanism (b).

694 **Fig. 8.** (a) The transmittance of the dissolution medium measured at 500 nm as a
695 function of time and (b) the sizes of the PL-DS particles measured by SDLC at the
696 end of the dissolution experiment.

697 **Fig. 9.** *Ex vivo* permeation profiles of the F2 fibers and pure DS (n=6). It should be
698 noted that it is not possible in the permeation experiment to distinguish between drug

699 in solution and in very small particles (< 220 nm) which could pass through the
700 filtration membrane used. Thus, some portion of the DS which had permeated in the
701 "free in water" experiment could in fact be in very small nanoparticles.

702 Fig. 10. The feasibility of different tri-axial electrospinning processes.

703

Table 1. Key details of the electrospinning processes and resultant fibers

No.	Process	F _O ^a (mL/h)	F _M ^a (mL/h)	F _I ^a (mL/h)	Morphology ^b	Diameter (μ m)
F1	Single	0	3.0	0	Linear	1.27 \pm 0.13
F2		0.5	2.0	0.5	Linear	0.55 \pm 0.06
F3	Tri-axial	0.5	1.6	0.9	Linear, with some beads	0.47 \pm 0.05
F4		1.0	1.6	0.4	Spindles-on-a-string	----

^a F_O, F_M and F_I are the flow rates of the outer, middle and inner fluids respectively.

^b “Linear” means the fibers have a straight-line morphology with few beads or spindles.